First-principles investigation of the structural, electronic and optical properties of Zn₂NbN₃

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An ab initio study using density functional theory (DFT) is carried out to explore the structural, electronic, and optical properties of Zn_2NbN_3 compound. The structural properties of this compound are determined by using the approximation (GGA-PBE) as implemented in WIEN2k. The calculated lattice parameters of the Zn_2NbN_3 compound are found to be a = 9.91 Å, b = 5.81 Å and c = 5.44 Å. The calculated electronic band structure and density of states indicate that the Zn_2NbN_3 compound is a wide gap semiconductor with a direct band gap of 2.5 eV. The different contributions of the electronic orbitals are discussed using the total and partial DOS with PBE and TB-mBJ approximations, which shows significant contributions from the Nb-d and N-p orbitals. The optical properties such as dielectric function, refractive index, absorption coefficient, and extinction coefficient are calculated and discussed.

Keywords: Zn₂NbN₃ ternary compound; Band structure; dielectric function; refractive index; density functional theory.

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1. Introduction

Since the 1990s, inorganic nitrides have attracted the interest of scientists for their chemical and physical properties beneficial for various industrial and technological fields [1]. Inorganic nitrides are divided into two categories according to their electronic properties: semiconductor wurtzite main group metal nitride and superconducting rocksalt transition metal (TM) nitride [2]. The first type such as AlN, GaN et InN present many important characteristics, including direct bandgap, high carrier mobility [1,3], they are largely used in laser diodes, solid states lighting, high data density optical storage media, radio frequency transistors [4,5]. The second type, such as TiN, VN and NbN, are employed in the hard coating, conduction barriers in semiconductor devices, and superconductors with transitions at a temperature around 20K [6,7]. Ternary nitride semiconductors are a large class of materials for a wide range of properties because of their unique nature and wide band gaps. Their intermediate bonding character due to the high energy of the nitrogen 2p orbital provides better hybridization with metal orbitals, making semiconductors crucial candidates in optoelectronic and photonic devices [8]. These materials have been widely studied in the past decade [9]. Principle calculations showed that various ternary zinc

nitrides exhibit electronic and optical properties. Using DFT calculations Yoyo Hinuma *et al.* 2016, investigated the electronic structure, stability and dopability of $CaZn_2N_3$ demonstrated tunable direct bandgap, light emission and harvesting. They have also proposed 21 interesting nitride systems, such as Ba_2ZnN_2 and Zn_2PN_3 [10]. Furthermore, M. Mallmann *et al.* (2018) have synthesized Mg₂PN₃ and Zn₂PN₃ ammonothermally [11] for the first time, and de-

termined bandgap based on diffuse reflectance spectroscopy [12], which estimated to be 5.0, 3.7 eV for Mg_2PN_3 and Zn₂PN₃ respectively. Afterwards, they performed DFT calculation to verify experimental values; both compounds crystallize in wurtzite (WZ)-type superstructure in orthorhombic space group Cmc2₁ [13]. In addition, Zakutayev et al. 2021 have recently synthesized the Zn₂NbN₃ compound for the first time. Experimental results indicate that Zn₂NbN₃ crystallizes in cation-disordered WZ-derived cristal structure with 2.1 eV optical band gap and c/a = 1.55 lattice constants [14]. In the same work, this group proved by theoritecal predictions an ordred structure of Zn₂NbN₃ with $Cmc2_1$ (space group 36) similar to Zn_2PN_3 and they demonstrated that it is energetically stable (formation energy equal to -0.52 eV/atom [15]. In this work, we realized density function theory study as yet unreported to investigate structural, electronic and optical properties of Zn₂NbN₃ nitrides. The paper is organized as follow. In Sec. 2 the computational methods and the relevant technical details are described. Section 3 presents our results and discussion. Finally, a short summary and conclusion are given in Sec. 4.

2. Computational method

The calculations were performed via first-principles fullpotential linearized augmented plane-wave (FP-LAPW) method [16] in the framework DFT calculations as implemented in WIEN2K code [17,18]. For the exchangecorrelation energy functional, there are two approximations, the local density approximation (LDA) and the generalized gradient approximation (GGA). Realized that the LDA approximation is the most basic non-trivial approximation to the exchangecorrelation energy. We assume that the energy XC of the system at a given point is equal to the energy of a homogeneous electron gas with (approximately) the same density at that point. However, we run into problems when describing systems with widely varying electron densities [19,20]. In our calculation, we used the generalized gradient approximation (GGA) in the Perdew-Burke-Ernzerhof (PBE) form instead of the LDA. Therefore, GGA is a class of functions that consider the density at a given point and the density's gradient (variation), allowing it to describe systems (to some extent) better. GGA has become a common denominator in many studies, especially when describing large-scale systems with different atoms [21]. In addition, the GGA-PBE approximation provides accurate results for the structural and electronic properties of semiconductors [22,23]. For the band gap calculations with accuracy, we perform GGA with the addition of Tran-Blaha modified Becke-Johnson (TB-mBJ) exchange potential [22-25]. The details of the calculations are as follow: The muffin tin radius sphere (R_{mt}) for each atom Zn (2.05), Nb (1.90) and N (1.63). All the crystal structure and the atomic coordinates are fully relaxed using the PORT method until the force on each atom converges to less than two mRy/u.a. The maximum angular momentum of the atomic orbital basis functions was set to $l_{\text{max}} = 6$. We expand the basis function up to $R_{mt} \times K_{max} = 7$, where Rmt denotes the smallest atomic sphere radius and K_{max} denotes the largest k vector in the plane wave expansion. The Fourier expansion charge density is truncated at $G_{\text{max}} = 12Ry^{1/2}$. The total number of k-points in the first Brillouin zone (BZ) was determined equal to 1000, and the k-point mesh was $10 \times 10 \times 9$. The iteration process stopped at a total energy convergence threshold unless 0.01 mRy.

3. Results and discussion

3.1. Structural properties

In our work, we report Zn_2NbN_3 wurtzite derived orthorhombic structure with $Cmc2_1$ (space group 36). In order to obtain the most stable structure, we optimized the structure by calculating different energies for different volumes using GGA-PBE adapting to the Murnaghan equation Eq. (1). The E(V)diagram is shown in Fig. 1b). The stable Zn_2NbN_3 wurtzite derived orthorhombic crystal with the ordered occupation of Zn and Nb by cationic sites is shown in Fig. 1a) and relaxed



FIGURE 1. Zn_2NbN_3 structure optimization a) Zn_2NbN_3 relaxed structure with atom's color are Zn (magenta), Nb (green), and N (blue) and b) diagram of energy versus volume.

lattice parameters are grouped in Table I as well as the previous theoretical and experimental results.

$$E(V) = E_0 + K_0 V_0 \left[\frac{1}{K'_0(K'_0 - 1)} \left(\frac{V}{V_0} \right)^{1 - K'_0} + \frac{1}{K'_0} \frac{V}{V_0} - \frac{1}{K'_0 - 1} \right].$$
 (1)

The thermodynamic stability of Zn_2NbN_3 was studied by the formation energy, which is described as the difference between the total energy of final products and these of reactants [26,27]. After relaxation, high-precision static calculations were performed to calculate the precise ground state energies of the reactants and product in order to compute the formation energy per atom of Zn_2NbN_3 using the following equation:

$$\Delta E_f = E_{\text{Zn}_2\text{NbN}_3}^{\text{tot}} - \alpha E(\text{Zn}) - \beta E(\text{Nb}) - \frac{\gamma}{2}E(\text{N}_2), \quad (2)$$

Lattice parameters	PBE	Other Theoretical works[14]	Experiment work [14]
space group	$Cmc2_1$	$Cmc2_1$	P63mc
crystal system	orthorhombic	orthorhombic	Hexagonal
$a(\text{\AA})$	9.91	9.852	3.36
$b(\text{\AA})$	5.81	5.782	3.36
$c(\text{\AA})$	5.44	5.386	5.26
Cation's ordre	ordered	ordered	disordered

TABLE I. Relaxed lattice parameters of Zn₂NbN₃.

where $E_{Zn_2NbN_3}^{tot}$ is the total energy per formula unit of the compound Zn₂NbN₃ under the equilibrium lattice constant, and E(Zn), E(Nb), and E(N2) are the ground-state energies of zinc (space group: 225 (Fm-3m)), niobium (space group: 225 (Fm-3m)), and nitrogen dimers (space group (225 (Fm-3m)), respectively. Generally, the better the structural stability, the more negative the formation energy value. In our case, the formation energy of Zn₂NbN₃ is -0.50 eV/atom, which means that our compound under study is thermodynamically stable; this value confirms other theoretical studies (DEf (Zn₂NbN₃) = -0.52 eV/atom) [15].

3.2. Electronic properties

The gap is one of the most important electronic parameters that allow us to get very useful informations about the nature, efficiency, usefulness of the material [22]. The exact bandgap of semiconductors are very appreciable for their efficient use in optoelectronic and other electronic devices, and minimal variation of its value induces significant changes in application devices [28]. In our study's calculations, the electronic bandgap of Zn_2NbN_3 is determined using (PBE) and (TB-mBJ). The results are shown in Table II.

Figure 2 illustrates band structure of Zn_2NbN_3 . We noticed that it is characterized by a direct bandgap with clear semiconducting behavior. Valence band is close to Fermi level, while conduction band do not. We can see that the maximum of valence band and the minimum of conduction band are located at the H point with a value of bandgap about 2 eV and 2.5 eV using (GGA-PBE) and (TB-mBJ) approximations, respectively.

On the other hand, the total density of states (DOS) and the partial DOS play an essential role in investigating the electronic properties of the compounds. In Figs. 3 and 4 show



FIGURE 2. Band structure of Zn_2NbN_3 compound, a)GGA-PBE approximation and b)TB-mBJ approximation.



FIGURE 3. The total density of States of Zn_2NbN_3 compound with both approximations GGA-PBE and TB-mBJ. a) GGA-PBE approximation, and b) TB-mBJ approximation.

the total DOS and partial DOS of Zn₂NbN₃ structure and Zn, Nb, and N atoms, respectively, calculated with the two potentials PBE and TB-mBJ. As shown in Fig. 3, the total DOS plot shows a zero-gap at the Fermi level with the two used potentials, confirming the semiconducting behavior in the electronic band structure plot. The total DOS simulated with PBE and TB-mBJ potentials is almost the character of the Zn atom, especially of d orbital in the energy range between -4.5 and -6 eV in the valence band [Fig. 4a) and 4d)]. The energy range -4.5 to 0 eV is dominated by Znd, Nb-d, and N-p orbitals [Figs. 4a) to 4f) respectively], indicating a strong interaction between these atoms. We can also see at the energy interval close to the Fermi level that the main contribution comes from the N-p orbital [Fig. 4c) and 4f)]. In the conduction band, the DFT calculations with the two potentials PBE and TB-mBJ indicate that the main contribution is from Nb-d and N-p orbitals [Figs. 4b), 4c), 4e) and 4f) respectively] with the domination of Nb-d orbital indicating a strong interaction between Nb and N atoms.

3.3. Optical properties

The optical properties of a material describe its response to electromagnetic radiation. A detailed optical characterization is thus conducted to determine the optoelectronic nature of



FIGURE 4. Total and partial DOS with PBE and TB-mBJ approximations of Zn a), d), Nb b), e) and N c), f), atoms respectivelly.

of Zn₂NbN₃, using the TB-mBJ approximation and thus support their use in the fabrication of optoelectronic devices.

3.3.1. Dielectric function

The optical properties of any material explain the frequency response to the energy of the incident photon ($E = h\nu$). Because the frequency depends on the dielectric function, we can use the latter to determine different optical parameters using the relation that expresses complex dielectric function [29]:

$$\varepsilon(\omega) = \varepsilon_1(\omega) + i\varepsilon_2(\omega), \tag{3}$$

where $\varepsilon_1(\omega)$ is the real part that helps to determine the polarization and dispersion of light, while $\varepsilon_2(\omega)$ is the imaginary part that describes the absorptive properties of the material. The two parts of the complex dielectric function $\varepsilon_1(\omega)$ and $\varepsilon_2(\omega)$ are related by the Kramers - Kronig relation. Optical properties depend strongly on the imaginary part of dielectric function $\varepsilon_2(\omega)$, which can be calculated using the following relation [30].

$$\varepsilon_{2}(\omega) = \frac{8\pi^{2}e^{2}}{\omega^{2}m^{2}} \sum_{n} \sum_{n'} \int |P_{nn'}^{v}(k)^{2}| f_{k_{n}}(1 - f_{k_{n}'})$$
$$\times \delta(E_{n}^{k} - E_{n'}^{k} - \hbar\omega) \frac{dk^{3}}{(2\pi)^{3}}, \qquad (4)$$

where $P_{nn'}^{v}(k)$ is the projection of elements of momentum dipole matrix along direction v of electric field between final and initial stages. f_{k_n} is Fermi-Dirac distribution, and E_n^k is the energy of one electron.

The Kramers-Kronig method can help us to extract the expression of the real part of the dielectric function $\varepsilon_1(\omega)$ using the imaginary part of dielectric function $\varepsilon_2(\omega)$ [31].

$$\varepsilon_1(\omega) = 1 + \frac{2}{\pi} P \int_0^\infty \frac{\omega' \varepsilon_2(\omega')}{\omega'^2 - \omega^2} d\omega'.$$
 (5)

Optical properties were computed using dielectric coefficient values. The real part of the dielectric function shows the ability of a material to respond to the incident electromagnetic wave. The knowledge of both real and imaginary parts of the dielectric function allows the calculation of important optical functions; the static value of real part $\varepsilon_1(0)$ is obtained on the vertical axis of the plot shown in Fig. 5 approximately 6.1 in xx direction 5.7 in zz direction. The real part of the dielectric function becomes negative at approximately 6.9 and 9.5 with high incident energy having anisotropy between xx and zz directions which shows reflectivity for photons for our considered compound Zn₂NbN₃. We can characterize



FIGURE 5. Real part of dielectric function of Zn₂NbN₃.



FIGURE 6. Imaginary part of dielectric function of Zn₂NbN₃.

electronic properties with the help of an imaginary part of dielectric function related to absorption phenomenon [32], the peaks in the spectra of the imaginary part of dielectric function describe transitions of electrons from the valence to conduction band [33]. Calculated spectra of imaginary part of dielectric function $\varepsilon_2(\omega)$ for Zn₂NbN₃ is shown in Fig. 6, the maximum value of $\varepsilon_2(\omega)$ located at 4 eV. The electronic band structure shows a direct bandgap; therefore, direct optical transitions occur between universal localized in the highest valence band (VBM) and the lowest conduction band (CBM). the main peaks in $\varepsilon_2(\omega)$ are due to interband electronic transitions from occupied N- 2p states localized in the VBM to the unoccupied Nb-4d states localized in the CBM (N-Nb bond) [Figs. 4e), 4f)]. It can be seen in the xx direction three intense peaks close to 9 in the UV region (5-7 eV); however, in the zz direction, there is one intense peak (7) at 5.5 eV; all of these features are attributed to interband transitions from occupied Zn-3d states localized in the highest valence band to the unoccupied Nb - 4d states localized in the lowest conduction band (Zn-Nb bond) Figs. 4d), 4f).

3.3.2. Absorption coefficient

The reason for a transition of an electron from the valence band to the conduction band is the interaction between incident photon and electron, which is related to absorption coefficient [34]. Figure 7 shows spectra of absorption coefficient. Calculations show that our compound possesses transparent behavior below the energy band gap (valance band) because the incident photon's energy is insufficient to transport electrons from the valance band to the conduction band. The first peak in the plot occurs at 4.04 eV with a value of optical gap estimated to be 3.1 eV. The second appears at 7.3 eV. After that, we notice a growth of the absorption with high energies indicating an effective optical behavior [33]. The appearance of the absorption beyond the energy 3.1 eV means that the absorption in semiconductors occurs when the incident energy exceeds the energy gap. In addition, strong absorption in the ultraviolet region suggests that Zn2NbN3 could be used in UV light protection [35].



FIGURE 7. Absorption coefficient plot of Zn₂NbN₃.



FIGURE 8. Refractive index plot of Zn₂NbN₃.

3.3.3. Refractive index

Refractive index can be calculated using the relation bellow with the help of real and imaginary part of dielectric function $\varepsilon_1(\omega)$ and $\varepsilon_2(\omega)$ [29].

$$n(\omega) = \sqrt{\frac{\sqrt{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)}}{2}} + \frac{\varepsilon_1(\omega)}{2}.$$
 (6)

The plot of the refractive index of Zn_2NbN_3 is shown in Fig. 8, the value of static refractive index n(0) is 2.4 and 2.5 in zz direction and xx direction, respectively, indicating the semiconducting behavior of the compound. The main peak has a maximum value at 3.5 eV energy value, so Zn_2NbN_3 has a high refractive index, proving that our compound can be used in photonic applications. After 12 eV, the refractive index decreases when the energy of the incident photon increases and the value of $n(\omega)$ becomes less than one, providing the photoluminescence phenomenon, which means that the phase velocity of passing light through the material is greater than the light speed in a vacuum [33]. We can also



FIGURE 9. Extinction coefficient $K(\omega)$.

observe a clear match between the plot of real part of dielectric function $\varepsilon_1(\omega)$ and the plot of refractive index $n(\omega)$ Fig. 5 and Fig. 8. The first peak in both plots is related to photons with minimal energy. After this peak, we notice a continuous decrease in both coefficients because of fluctuations.

3.3.4. Extinction coefficient

The extinction coefficient shows the amount of radiation absorbed by the material, which explains the similarity between the extinction coefficient and absorption coefficient plots. Extinction coefficient $K(\omega)$ can be also calculated using the values of $\varepsilon_1(\omega)$ and $\varepsilon_2(\omega)$ according to following relation [32].

$$K(\omega) = \sqrt{\frac{\sqrt{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)}}{2}} - \frac{\varepsilon_1(\omega)}{2}.$$
 (7)

Figure 9 shows extinction coefficient spectra. We can see the extinction coefficient close to zero in the energy range from 0.0 to 2.0 eV, which means that the electromagnetic wave can completely pass through the material. The first peak is of the order of 1.5, around 4.0 eV. An important peak of 2.2 occurs at 7.0 eV. Hence, the electromagnetic wave passes difficulty through the material after the energy of 2.0 eV.

4. Conclusion

We have computed structural, electronic, and optical properties for Zn₂NbN₃ by applying DFT theory-based FP-LAPW approach. Structural parameters are a = 9.90 Å and b =5.81 Å and c = 5.43 Å. The electronic calculations obtained from band structure using both (GGA) and (TB-mBJ) approximations prove direct bandgap and semiconductor behavior. From total and partial DOS performed by employing (GGA) and TB-mBJ approximations, we have concluded that the main contribution is from Nb-d and N-p orbitals, in good agreement with the imaginary part of the dielectric function. The absorption of Zn₂NbN₃ appears after 3.1 eV, consistent with results from electronic properties, using (PBE) and (TBmBJ) approximations. The Zn₂NbN₃ material has a high refractive index and strong optical absorption in the ultraviolet region. So, this compound is suitable for optoelectronic applications.

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