

Theoretical studies on the g-factors and the local structure of W^{5+} ions in tungsten phosphate glasses

B. F. Zhang^a, J. Z. Lin^{b,*}, Y. X. Wang^b, Y. Zhao^b and Y. F. Zhang^b

*Department of electronic information engineering, Suzhou Institute of Technology,
Jiangsu university of science and technology, Zhangjiagang, China 215600.*

*Department of physics, Suzhou Institute of Technology,
Jiangsu university of science and technology, Zhangjiagang, China 215600.*

**e-mail: linjizi@just.edu.cn*

Received 05 June 2022; accepted 05 January 2023

In this work, we adopt the three-order perturbation formulae for g-factors (g_{\parallel} , g_{\perp}) of d^1 ions in the octahedral environment to calculate the g-factors of W^{5+} ions in tungsten phosphate glasses containing lithium ($P_2O_5-Li_2WO_4-Li_2O$). In the light of the high valence state of the studied W^{5+} centers and hence the strong covalency of the studied octahedral $[WO_6]^{7-}$ cluster, we consider the contributions to g-factors from the ligand orbital and spin-orbit (SO) coupling interactions based on the cluster approach. The required tetragonal crystal-field parameters are calculated from the local structure of W^{5+} ions based on the superposition model. According to the theoretical calculations, we find that the octahedral $[WO_6]^{7-}$ clusters possess the tetragonally compressed distortion with a shorter W-O bond length ($\approx 1.54 \text{ \AA}$) and a longer one ($\approx 2.26 \text{ \AA}$) along C_4 axis and four normal W-O bond length ($\approx 1.94 \text{ \AA}$) in the perpendicular plane, which infers that the W^{5+} ions are in the form of tungstyl ions (*i.e.*, WO^{3+}). Based on the local structural data, the theoretical values of g_{\parallel} and g_{\perp} agree well with the experimental values.

Keywords: g-factors; W^{5+} ; tungsten phosphate glasses; superposition model; cluster approach.

DOI: <https://doi.org/10.31349/RevMexFis.69.040402>

1. Introduction

Glass doped with transition metal ions has many excellent properties (*e.g.* high refractive index [1–3], low phonon energy [4, 5], unique non-linear optical properties [6, 7]), which has attracted great interest of many researchers [8]. Many theoretical and experimental studies show that the different structures of oxygen coordination polyhedron in glass network strongly correlated with the excellent properties of the glasses system [8–10]. As it is known, one can adopt the electron paramagnetic resonance (EPR) technique to conveniently study the structures of the glasses by using probing agents such as the transition metal ions (*e.g.* W^{5+} [8, 11–13], Mo^{5+} [14–16], VO^{2+} [17–20], Cu^{2+} [17, 21–23], Cr^{5+} [24–26]). EPR experiments for the glasses system $P_2O_5-Li_2WO_4-Li_2O$ were carried out decades ago [15] and the EPR parameters g-factors (g_{\parallel} and g_{\perp}) of the W^{5+} ions reduced from W^{6+} in the formation of the glasses system $P_2O_5-Li_2WO_4-Li_2O$ were obtained in the experiment [15]. From the property of the experimental g-factors (*i.e.*, $g_{\parallel} < g_{\perp} < g_e$, $g_e \approx 2.0023$ is the spin-only value), the W^{5+} ion was found to locate at the axially distorted octahedral environment, in which the metallic ion W^{5+} was in the form of tungstyl ion WO^{3+} with a short W-O bond length [15]. However, up to date, the g-factors of W^{5+} ions in the glasses system $P_2O_5-Li_2WO_4-Li_2O$ have not been satisfactorily interpreted and the local structure of the metallic ions W^{5+} in the system has not been confirmed in the quantitative calculation. Considering that the microscopic mechanisms

of the EPR parameters and the local structure of the metallic ion W^{5+} must be essential to understand the properties of the $P_2O_5-Li_2WO_4-Li_2O$ glasses system, further theoretical investigations on the experimental results of the system are of great significance.

Here, we apply the three-order perturbation formulae of g-factors for the d^1 ions in the octahedral environment to theoretically investigate these factors of the studied W^{5+} ($5d^1$) ions. Because of the high valence state of the studied ion, one can expect significant orbital admixtures between the central ions and the ligand ions. Thus, the contributions to g-factors from the ligand orbital and spin-orbit (SO) coupling interactions are included in the perturbation formulae based on the cluster approach. According to the superposition model (SPM), the required crystal-field energy separations in these formulae are calculated from the parameter of the local environment around the studied W^{5+} ions. Thus, information on the local structure of the metallic ion is confirmed. The theoretical results (g_{\parallel} and g_{\perp}) present here show good agreement with the experimental results.

2. Calculations

2.1. The local structure and energy level splitting for W^{5+} in the $P_2O_5-Li_2WO_4-Li_2O$ glasses system

In Ref. [15], the obtained EPR spectra characterized by the major signal can be ascribed to the W^{5+} ion reduced from the W^{6+} ion in the formation of the glasses system. Based

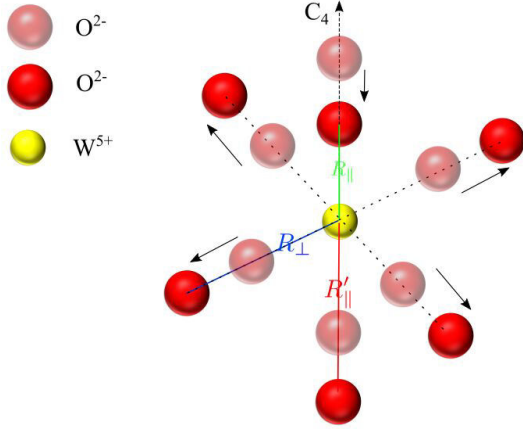


FIGURE 1. (Color online) Local structure of the octahedral $[\text{WO}_6]^{7-}$ cluster with tetragonally compressed distortion in $\text{P}_2\text{O}_5\text{-Li}_2\text{WO}_4\text{-Li}_2\text{O}$ glasses. The W^{5+} ions are in the form of tungstly ions WO^{3+} with a short W-O bond length R_{\parallel} (described by the green solid line) and a long one R'_{\parallel} (shown by red solid line) along the C_4 axis and four normal perpendicular W-O bond length R_{\perp} (shown by the blue solid line).

on the feature $g_{\parallel} < g_{\perp} < g_e$ of the experimental g-factors, one can assume that these W^{5+} ions are coordinated by six oxygen ligands forming the octahedral environment. As observed in other glasses [8, 12, 13], the metallic ion W^{5+} should be very close to one of the two oxygen ligands along the C_4 axis leading to the axially compressed distortion octahedron (see Fig. 1). Therefore, the shorter axial bond length R_{\parallel} and the longer axial bond length R'_{\parallel} as well as the four normal W-O bond length R_{\perp} in the perpendicular plane can be expressed by the reference length R and the relative tetragonal compression ratio ρ :

$$\begin{aligned} R_{\parallel} &\approx R(1 - \rho), & R'_{\parallel} &\approx R(1 + \rho), \\ R_{\perp} &\approx R(1 + \rho^2/2). \end{aligned} \quad (1)$$

For the free d^1 ions, the d-orbital is five-fold degenerate. In cubic crystal field, the quintuple degeneracy is partially relieved and the octahedral d^1 cluster exhibits two energy levels 2E_g ($|d_{z^2}\rangle, |d_{x^2-y^2}\rangle$) and ${}^2T_{2g}$ ($|d_{xy}\rangle, |d_{yz}\rangle, |d_{xz}\rangle$), as shown in Fig. 2. The original lower orbital triplet ${}^2T_{2g}$ in cubic crystal-field would split into an orbital doublet 2E_g ($|d_{yz}\rangle, |d_{xz}\rangle$) and an orbital single ${}^2B_{2g}$ ($|d_{xy}\rangle$) due to the tetragonally compressed distortion with the latter being lowest [27]. Meanwhile, the original upper orbital doublet 2E_g would split into two orbital singlets ${}^2B_{1g}$ ($|d_{x^2-y^2}\rangle$) and ${}^2A_{1g}$ ($|d_{z^2}\rangle$). According to crystal field theory [27], the Hamiltonian matrices for d^1 ion in the tetragonally compressed octahedron can be expressed by the cubic crystal-field parameter (CF) D_q , the tetragonal CF parameters D_s and D_t ,

$$\begin{aligned} \langle d_{z^2} | V | d_{z^2} \rangle &= 6D_q - 2D_s - 6D_t, \\ \langle d_{x^2-y^2} | V | d_{x^2-y^2} \rangle &= 6D_q + 2D_s - D_t, \end{aligned}$$

FIGURE 2. (Color online) Crystal-field energy splitting of d^1 ion in an octahedral environment with the tetragonally compressed distortion.

$$\begin{aligned} \langle d_{xz} | V | d_{xz} \rangle &= \langle d_{yz} | V | d_{yz} \rangle = -4D_q - D_s + 4D_t, \\ \langle d_{xy} | V | d_{xy} \rangle &= -4D_q + 2D_s - D_t. \end{aligned} \quad (2)$$

Here, V denotes the crystal field potential energy of the d^1 electron. Thus, the energy differences E_i ($i=1,2,3$, see Fig. 2) between the excited 2E_g , ${}^2B_{1g}$, ${}^2A_{1g}$ and the ground state ${}^2B_{2g}$ can be calculated [28–30]:

$$\begin{aligned} E_1 &= 5D_t - 3D_s, & E_2 &= 10D_q, \\ E_3 &= 10D_q - 4D_s - 5D_t. \end{aligned} \quad (3)$$

According to the superposition model [31] and the local geometrical relationship (see Fig. 1), the tetragonal CF parameters D_s and D_t can be written as follows:

$$\begin{aligned} D_s &= (2/7)\bar{A}_2(R)[2(R/R_{\perp})^{t_2} - (R/R_{\parallel})^{t_2} - (R/R'_{\parallel})^{t_2}], \\ D_t &= -(8/21)\bar{A}_4(R)[2(R/R_{\perp})^{t_4} \\ &\quad - (R/R_{\parallel})^{t_4} - (R/R'_{\parallel})^{t_4}], \end{aligned} \quad (4)$$

where $\bar{A}_2(R)$ and $\bar{A}_4(R)$ are the intrinsic parameters with the reference length R [31]. The ratios $\bar{A}_4(R)/D_q \approx 3/4$, $\bar{A}_2(R)/\bar{A}_4(R) \approx 10.8$ have been proved to be valid for d^n ions in octahedral crystal-field environments for many glasses systems [9,32,33]. They can be reasonably applied in the present work. t_2 (≈ 3) and t_4 (≈ 5) are the power-law exponents [31]. According to the optical absorption spectra of the similar $[\text{WO}_6]^{7-}$ cluster in some tungstate glasses [13], one can obtain $D_q \approx 2500 \text{ cm}^{-1}$. Thus, only the parameter ρ is unknown in Eq. (4).

2.2. Perturbation formulae of g-factors

In the light of the high valence state of the metallic W^{5+} ion in $[\text{WO}_6]^{7-}$ cluster, one can expect the strong covalency effect and hence the obvious mixings between the orbitals of the ligand O^{2-} and the central W^{5+} ions. Thus, the contributions to EPR parameters (g_{\parallel} , g_{\perp}) from the ligand orbital and SO coupling interactions become significant, which has been proved in many theoretical works [13]. For the W^{5+} ion in the tetragonally compressed octahedral environment, the high-order perturbation formulae of g-factors including

TABLE I. (ζ , ζ') (in cm⁻¹), N_γ , (λ_γ , λ_s) and (k , k') for W⁵⁺ ion in P₂O₅-Li₂WO₄-Li₂O glasses.

N_t	N_e	λ_t	λ_e	λ_s	ζ	ζ'	k	k'
0.6708	0.7180	0.7334	0.5878	0.4720	2457	2339	0.8512	0.3898

the above contributions can be derived on the basis of the cluster approach [34–36]:

$$\begin{aligned}
g_{\parallel} &= g_e - 8k'\zeta'/E_2 - 2k\zeta^2/E_1^2 - 4k'\zeta\zeta'/E_1E_2 - 2\zeta^2/E_1^2, \\
g_{\perp} &= g_e - 2k\zeta/E_1 + k\zeta^2/E_1^2 - 2k\zeta'^2/E_1E_2 \\
&\quad + 2k'\zeta'\zeta/E_1E_2 - 4\zeta'/E_2^2 - \zeta^2/E_1^2. \quad (5)
\end{aligned}$$

Here, energy separations E_1 and E_2 corresponding to d-d transitions are given in Eq. (3). ζ and ζ' are the SO coupling coefficients, k and k' are the orbital reduction factors characterizing the degree of covalency between the central W⁵⁺ and the ligands O²⁻ in present glasses system. These quantities (ζ , ζ' , k , k') can be derived from the cluster approach [9, 35]:

$$\begin{aligned}
\zeta &= N_t (\zeta_d + \lambda_t^2 \zeta_p / 2), \\
\zeta' &= (N_t N_e)^{1/2} (\zeta_d - \lambda_t \lambda_e \zeta_p / 2), \\
k &= N_t (1 + \lambda_t^2 / 2), \\
k' &= (N_t N_e)^{1/2} [1 - \lambda_t (\lambda_e + \lambda_s A) / 2]. \quad (6)
\end{aligned}$$

The subscript $\gamma = e$ and t denote the irreducible representations T_{2g} and E_g of the O_h group, respectively. N_γ is the normalization factor. λ_γ or (λ_s) is the orbital mixing coefficients. ζ_d and ζ_p are the SO coupling coefficients of the metal and ligand ions in free state, respectively. Based on the cluster approach, these molecular orbital parameters are calculated by using the covalency factor N , the group overlap integrals (S_{dpt} , S_{dpe} , S_{ds}) and the integral A . Here, the overlap integrals S_{dpt} and S_{dpe} involve the W(d) orbital and the O(2p) orbitals with the same T_{2g} and E_g symmetries, respectively. Also, S_{ds} involves the W(d) and the O(2s) orbitals with the same A_{1g} symmetry. A denotes the integral $R \langle ns | \partial/\partial y | np_y \rangle$ with the impurity-ligand distance R of the studied system. More information can be found in the literatures [28, 35, 37, 38].

2.3. Calculated results of the g-factors for W⁵⁺ ions in P₂O₅-Li₂WO₄-Li₂O glasses system

For the octahedral cluster [WO₆]⁷⁻ in some tungsten phosphate glasses [12], the average bond length is about 1.9 Å, which can be reasonably taken as the reference length in this work. Utilizing the length R (≈ 1.9 Å [12]) and the Slater-type SCF functions, the group overlap integrals $S_{dpt} \approx 0.03211$, $S_{dpe} \approx 0.09075$, $S_{ds} \approx 0.07287$ and the integral $A \approx 1.2371$ are determined. From the free-ion values $\zeta_d \approx 3500$ cm⁻¹ [36, 39] for W⁵⁺ and $\zeta_p \approx 151$ cm⁻¹ [9, 28] for O²⁻, the SO coupling coefficients (ζ , ζ') and the

TABLE II. The g-factors for W⁵⁺ ion in P₂O₅-Li₂WO₄-Li₂O glasses.

	Cal ^a	Cal ^b	Cal ^c	Exp ^d
g_{\parallel}	1.4142	1.7106	1.5949	1.5932-1.6015
g_{\perp}	1.8114	1.7331	1.7321	1.7321-1.7398

^aTheoretical results based on Eqs. (5)-(7) but neglecting the ligand contributions. ^bTheoretical results based on Eqs. (6), (7) and the second-order perturbation formulae (*i.e.*, the formulae of g-factors include only the first two terms in Eq. (5)). ^cTheoretical results based on Eqs. (5)-(7) including the ligand contributions. ^dRef. [15].

orbital reduction factors (k , k') are determined as the covalency factor N is known. By inserting the relevant quantities (E_1 , E_2 , ζ , ζ') into the formulae of g-factors (Eq. (5)) and fitting the theoretical g-factors to the experimental results, N and ρ are determined:

$$N \approx 0.655; \quad \rho \approx 0.1915. \quad (7)$$

The corresponding calculated results (N_γ , λ_γ , λ_s , ζ , ζ' , k , k') are collected in Table I and the corresponding g-factors (g_{\parallel} , g_{\perp}) are gathered in Table II. For the convenience of comparison and discussion, the calculated results (Cal^a) based on the relative compression ratio ρ without the ligand orbital and SO coupling contributions (*i.e.*, $\zeta = \zeta' = N\zeta_d$ and $k = k' = N$) are collected in Table II. The calculated results (Cal^b) based on the second-order perturbation formulae including the ligand contributions are also shown in Table II.

3. Discussion

From Table II, one can find that the theoretical g-factors (Cal^c) based on the three-order perturbation formulae of these factors for d^1 ions in the tetragonally compressed octahedral environment including the contributions from the ligand orbital and SO coupling interactions as well as the relative tetragonal compression ratio ρ are in good agreement with the experimental results reported in Ref. [15]. Therefore, the experimental EPR spectra are satisfactorily explained and the local structure is quantitatively confirmed.

- (1) Based on the relative compression ratio ρ (≈ 0.1915) and the reference length R (≈ 1.9 Å [15]), one can obtain the local structural data for the studied cluster [WO₆]⁷⁻ including the very short W-O bond length R_{\parallel} (≈ 1.54 Å) and the long one R'_{\parallel} (≈ 2.26 Å) along the C₄ axis and the four normal perpendicular W-O bond length R_{\perp} (≈ 1.94 Å), which infers a tetragonal compression distortion along the C₄ axis (see Fig. 1). This

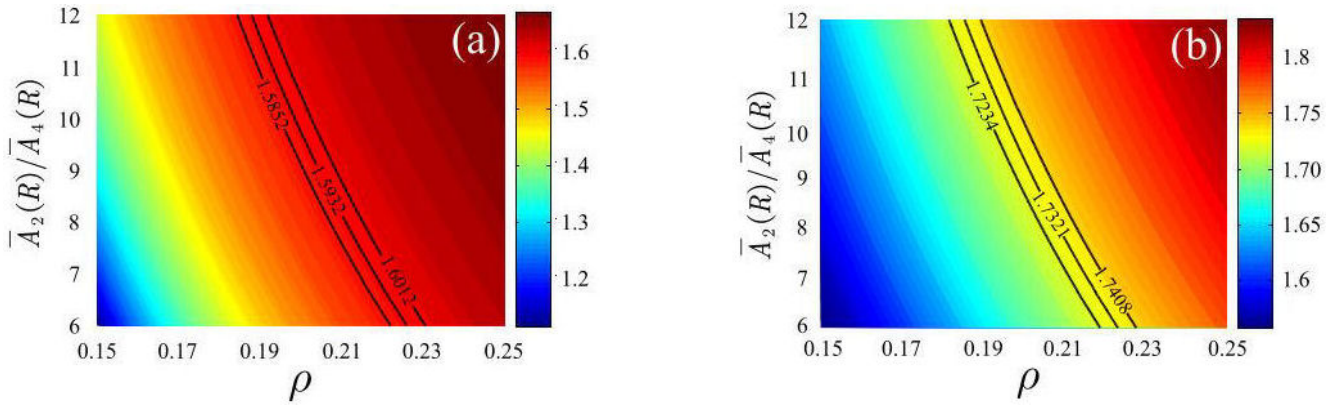


FIGURE 3. (Color online) g_{\parallel} and g_{\perp} as functions of the relative compression ratio ρ and the ratio $\bar{A}_2(R)/\bar{A}_4(R)$ are shown in Panels a) and b), respectively. In the two panels, the three black curves from left to right represent 99.5%, 100% and 100.5% of the corresponding experimental values, respectively.

may suitably explain the observed property ($g_{\parallel} < g_{\perp} < g_e$) of g-factors. Similar tetragonally compressed distortions of the octahedral W^{5+} centers are also found in other tungstate phosphate glasses (e.g. $R_{\perp} \approx 1.6\text{\AA}$, $R'_{\parallel} \approx 2.2\text{\AA}$, $R_{\perp} \approx 1.9\text{\AA}$) [12] by EPR analysis. Many theoretical works demonstrate that the similar distortions may be ascribed to the Jahn-Teller effect which may modify the immediate local environment by stretching or shrinking the metal-ligand bond lengths [8, 9, 28, 40, 41]. Thus, the local environment around the W^{5+} ion confirmed in this work is suitable.

- (2) It can be seen from Table II that the calculated g-factors (Cal^c) based on Eqs. (5), (6) and (7) are more consistent with the experimental results than the calculated results (Cal^a) based on Eqs. (5) and (7) without the contributions from the ligand orbital and spin-orbit coupling interactions. Particularly, the calculated value of g_{\parallel} from the results (Cal^a) is much smaller than the experimental value. According to our calculations, the relative discrepancy between the calculated results (Cal^a) and the experimental values are about 11.3% and 4.6% for g_{\parallel} and g_{\perp} respectively, which can hardly be eliminated by freely adjusting the relative tetragonal compression ratio ρ and the covalency factor N . In fact, one can expect the strong covalency and hence the obvious mixings between the metal and ligand orbitals because of the high valence state of W^{5+} . This point is supported by the small covalency factor N ($\approx 0.655 \ll 1$) and the significant mixing coefficients ($\lambda_t \approx 0.7334$, $\lambda_e \approx 0.5878$ and $\lambda_s \approx 0.4720$) (see Table I). By comparing the results between Cal^a and Cal^b in Table II, we also find that the contributions from the ligand orbital and SO coupling interactions are more important than those from the third-order terms. Therefore, the high-order perturbation formulae of the g-factors including the ligand contributions seem to be much applicable in the explanations of these parameters.

- (3) The possible errors of the present work may be analyzed in the following aspects. Firstly, only the contributions from the nearest neighbor oxygen ligands of the metallic W^{5+} ion are considered from the cluster approach. Fortunately, the approach has been proved to be valid by many theoretical works on EPR parameters [8, 9, 28, 29, 32–34, 38–40].

Secondly, the empirical relationship $\bar{A}_2(R)/\bar{A}_4(R) \approx 10.8$ applied in the calculations of tetragonal CF parameters may bring some errors to the compression ratio ρ and hence the g-factors (note that the relationship $\bar{A}_2(R)/\bar{A}_4(R) \approx 9 \sim 12$ for d^n ions has been proved to be valid in many theoretical works [29, 31, 33, 40, 42–47]). According to our calculations, the errors of the final g-factors and the relative compression ratio ρ are estimated not to exceed 1% and 6% (see Fig. 3) respectively, as the ratio $\bar{A}_2(R)/\bar{A}_4(R)$ changes by 10% around 10.8. Thirdly, in the calculations of the group overlap integrals, the reference bonding length of the studied $[WO_6]^{7-}$ cluster is obtained from the similar cluster with tetragonally compressed distortion in the other tungstate phosphate glasses. This would introduce small errors into the integrals and hence bring forward an influence in the calculated g-factors. However, because of the small integrals, one can expect the errors arising from the uncertainty of the integrals must be very small. According to the calculations, when the integrals change by 10%, the errors of the final g-factors are estimated not to exceed 1%. Finally, only the contributions from the crystal-field mechanism are considered in the calculations of the g-factors, while the contributions due to the charge-transfer mechanism are not included. This should lead to some errors in the final g-factors. However, considering that (i) some adjustable parameters (e.g. the bonding molecular orbit coefficients, charge-transfer energy levels [34, 36, 39, 47]) should be introduced in the calculations if the charge-transfer mechanism is adopted. (ii) Based on the theoretical studies of the octahedral clusters for the d^1 ions (e.g. V^{4+} [47, 48], Mo^{5+} [34, 49], W^{5+} [36, 39], Cr^{5+} [25, 47, 48]) doping in the crystals or glasses, the contri-

bution to the g-factors from the charge-transfer mechanism is much smaller than that from the crystal-field mechanism, neglecting the contribution to g-factors from the charge-transfer mechanism is acceptable. The theoretical g-factors (Cal^c) show good agreement with the experimental values, which further supports this point.

4. Conclusion

The g-factors (g_{\parallel} , g_{\perp}) and the local structure for tetragonal W⁵⁺ centers in tungsten phosphate glasses containing lithium (P₂O₅-Li₂WO₄-Li₂O) are quantitatively investigated through the three-order perturbation formulae for the d¹ ions in the octahedral environment. In these formulae, the contributions to g-factors from the ligand orbital and SO cou-

pling interactions are taken into account because of the strong covalency of the studied cluster [WO₆]⁷⁻. According to our calculations, the relative tetragonal compression ratio ρ is found to be about 0.1915 due to the Jahn-Teller effect. Based on the local structural data and the three-order perturbation formulae, the theoretical results of g-factors are in good agreement with the experimental results.

Acknowledgments

This work is partly supported by the National Natural Science Foundation of China (Grant No.12104191) and the Foundation of the College Natural Science Research Project of Jiangsu Province (Grant No.21KJB140006).

1. M. H. Zhang *et al.*, Study on novel high refractive index La₂O₃-Lu₂O₃-TiO₂ glasses prepared by aerodynamic levitation method, *Mater. Lett.*, **222** (2018) 5, <https://doi.org/10.1016/j.matlet.2018.03.120>
2. T. Z. Xu *et al.*, Anti-corrosion LaO₃/2-GaO₃/2-ZrO₂ infrared glasses with high refractive index and low dispersion prepared by aerodynamic levitation, *Opt. Mater.*, **114** (2021) 110943, <https://doi.org/10.1016/j.optmat.2021.110943>
3. K. S. Shaaban, M. Koubisy, H.Y. Zahran, and I. S. Yahia, Spectroscopic Properties, Electronic Polarizability, and Optical Basicity of Titanium-Cadmium Tellurite Glasses Doped with Different Amounts of Lanthanum, *J. Inorg. Organomet. P.*, **30** (2020) 4999, <https://doi.org/10.1007/s10904-020-01640-4>
4. A.E. Ersundu *et al.*, Characterization of new Sb₂O₃-based multicomponent heavy metal oxide glasses, *J. Alloy. Compd.*, **615** (2014) 712, <https://doi.org/10.1016/j.jallcom.2014.07.024>
5. Z. Wang *et al.*, Enhanced luminescence properties of Ho/Yb ions regulated by the nanocrystalline environment and phonon energy in silicate glasses, *J. Lumin.*, **219** (2020) 116949, <https://doi.org/10.1016/j.jlumin.2019.116949>
6. J. Rajagukguk, B. Sinaga, and J. Kaewkhao, Structural and spectroscopic properties of Er³⁺ doped sodium lithium borate glasses, *Spectrochim. Acta. A.*, **223** (2019) 117342, <https://doi.org/10.1016/j.saa.2019.117342>
7. S. N. Nazrin, M.K. Halimah, and F.D. Muhammad, Comparison study of optical properties on erbium-doped and silver-doped zinc tellurite glass system for non-linear application, *J. Mater. Sci-Mater. El.*, **30** (2019) 6378, <https://doi.org/10.1007/s10854-019-00940-x>
8. X. S. Liu *et al.*, Theoretical investigations on the defect structures and g factors for W⁵⁺ in lead bismuth arsenate glasses, *J. Non-Cryst. Solids*, **545** (2020) 120239, <https://doi.org/10.1016/j.jnoncrysol.2020.120239>
9. C. C. Ding *et al.*, Theoretical study of the local structures and the EPR parameters for RLNB glasses with VO²⁺ and Cu²⁺ dopants, *J. Phys. Chem. Solids*, **113** (2018) 102, <https://doi.org/10.1016/j.jpcs.2017.10.012>
10. H. Li, H. Lin, W. Chen, and L. Luo, IR and Raman investigation on the structure of (100-x)[0.33B₂O₃-0.67ZnO-xV₂O₅] glasses, *J. Non-Cryst. Solids*, **352** (2006) 3069, <https://doi.org/10.1016/j.jnoncrysol.2006.03.073>
11. N. S. Rao *et al.*, Investigation on spectral features of tungsten ions in PbO-Bi₂O₃-As₂O₃ glass matrix, *Physica B*, **406** (2011) 4494, <https://doi.org/10.1016/j.physb.2011.09.014>
12. F. Studer, N. Rih, and B. Raveau, Mixed valence tungsten phosphate glasses: ESR and diffuse reflectance investigation, *J. Non-Cryst. Solids*, **107** (1988) 101, [https://doi.org/10.1016/0022-3093\(88\)90098-1](https://doi.org/10.1016/0022-3093(88)90098-1)
13. P. N. Huang, X. H. Huang, and F. X. Gan, Optical absorption and ESR spectra of mixed conductive glasses in Li₂O-B₂O₃-WO₃ system, *Solid State Ionics*, **57** (1992) 169, [https://doi.org/10.1016/0167-2738\(92\)90080-9](https://doi.org/10.1016/0167-2738(92)90080-9)
14. O. Cozar, D. A. Magdas, and I. Ardelean, EPR study of molybdenum-lead-phosphate glasses, *J. Non-Cryst. Solids*, **354** (2008), 1032, <https://doi.org/10.1016/j.jnoncrysol.2007.08.002>
15. M. Haddad, A. Nadiri, A. Biyadi, M. E. Archidi, J. V. Folgado, and D. Beltran-Porter, Computer simulations of electron paramagnetic resonance spectra of P₂O₅ Li₂WO₄ Li₂O glasses, *J. Alloy. Compd.*, **188** (1992) 161, [https://doi.org/10.1016/0925-8388\(92\)90666-w](https://doi.org/10.1016/0925-8388(92)90666-w)
16. K. Neeraja, A. Rupesh Kumar, T. Rao, P. Vijayalakshmi, and M. Rami Reddy, The Spectroscopic Properties of Mo⁵⁺ in ZnO-ZnF₂-B₂O₃ Glasses, *T. Indian Ceram. Soc.*, **72** (2013) 24, <https://doi.org/10.1080/0371750X.2013.794001>
17. B. Srinivas, A. Hameed, G. Ramadevudu, M.N. Chary, and M. Shareefuddin, The Spectroscopic Properties of Mo⁵⁺ in ZnO-ZnF₂-B₂O₃ Glasses, *J. Phys. Chem. Solids*, **129** (2019) 22, <https://doi.org/10.1016/j.jpcs.2018.12.042>

18. B. Srinivas, A. Hameed, R. Vijaya Kumar, M. Narasimha Chary, and M. Shareefuddin, Evaluation of EPR parameters for compressed and elongated local structures of VO₂⁺ and Cu₂⁺ spin probes in BaO-TeO₂-B₂O₃ glasses, *Philos. Mag.* **98** (2018) 1625, <https://doi.org/10.1080/14786435.2018.1450530>
19. A. Rajiv, J. Uchil, and C. N. Reddy, EPR studies on VO₂⁺-doped sodium-borate glasses synthesized by microwave method, *IOP Conference Series: Materials Science and Engineering*, **360** (2018) 12040, <https://doi.org/10.1088/1757-899X/360/1/012040>
20. A. V. Lalithaphani, B. Srinivas, A. Hameed, M. N. Chary, and M. Shareefuddin, Electron paramagnetic resonance spectra of CdO-Al₂O₃-Bi₂O₃-B₂O₃ quaternary glasses containing VO₂⁺ ions, *AIP Conference Proceedings*, **1942** (2018) 70011, <https://doi.org/10.1063/1.5028809>
21. M. R. Ahmed and M. Shareefuddin, EPR, optical, physical and structural studies of strontium aluminoborate glasses containing Cu₂⁺ ions, *SN Applied Sciences* **1** (2019) 1, <https://doi.org/10.1007/s42452-019-0201-5>
22. D. F. Franco *et al.*, Structural and EPR studies of Cu₂⁺ ions in NaPO₃-Sb₂O₃-CuO glasses, *J. Non-Cryst. Solids*, **503** (2019) 169, <https://doi.org/10.1016/j.jnoncrysol.2018.09.042>
23. K. Biramally, N. Pujari, S. Karidas, M. D. Shareefuddin, D. Sreenivasu, and C. P. Vardhani, Electron paramagnetic resonance and optical absorption studies of Cu₂⁺ probe in Li₂O-Bi₂O₃-As₂O₃-B₂O₃ glasses, *AIP Conference Proceedings* **2104** (2019) 30016, <https://doi.org/10.1063/1.5100443>
24. R. P. S. Chakradhar, J. L. Rao, G. Sivaramaiah, and N.O. Gopal, Chromium ions in alkali lead borotellurite glasses - An EPR and optical study, *Phys. Status. Solidi. B.*, **242** (2005) 2919, <https://doi.org/10.1002/pssb.200540100>
25. Y. Mei, W. C. Zheng, B. W. Chen, and C. F. Wei, Investigations of the optical spectral bands, g factors and local structure for the tetragonal Cr⁵⁺ tetrahedral clusters in Cr-doped silica glasses, *J. Alloy. Compd.*, **614** (2014) 140, <https://doi.org/10.1016/j.jallcom.2014.06.069>
26. W. Stręk *et al.*, Spectroscopic studies of chromium-doped silica sol-gel glasses, *J. Non-Cryst. Solids*. **288** (2001) 56, [https://doi.org/10.1016/S0022-3093\(01\)00610-X](https://doi.org/10.1016/S0022-3093(01)00610-X)
27. A. Abragman, and B. Bleaney, Electron paramagnetic resonance of transition ions, (Oxford University Press, London, 1970).
28. H. M. Zhang, W. B. Xiao, and X. Wan, Theoretical studies of the optical and EPR spectra for VO₂⁺ in MgKPO₄·6H₂O single crystal, *Physica B* **449** (2014) 225, <https://doi.org/10.1016/j.physb.2014.05.036>
29. Q. Zhang, S. Y. Wu, Z. Y. Zhang, G. J. Chen, Q. S. Zhu, and X. Y. Li, Theoretical investigations on the EPR parameters and local structures for V⁴⁺ in alkali niobium borate glasses, *J. Non-Cryst. Solids* **522** (2019) 119559, <https://doi.org/10.1016/j.jnoncrysol.2019.119559>
30. W. Chen *et al.*, Investigation on the EPR Parameters for Tetragonal VO₂⁺ Centers in LiHSO₄ Single Crystals, *Iran. J. Sci. Technol. A.*, **44** (2020) 1559, <https://doi.org/10.1007/s40995-020-00947-8>
31. D. J. Newman and B. Ng, The superposition model of crystal fields, *Rep. Prog. Phys.*, **52** (1989) 699, <https://doi.org/10.1088/0034-4885/52/6/002>
32. Y. J. Hong, S. Y. Wu, J. R. Lyu, H. Wu, and H. N. Dong, Theoretical studies on the concentration dependences of optical spectra and spin Hamiltonian parameters for V⁴⁺ in CdO-SrO-B₂O₃-SiO₂ glasses, *J. Phys. Chem. Solids.*, **161** (2022) 110480, <https://doi.org/10.1016/j.jpcs.2021.110480>
33. W. Jin, X. H. Chu, C. C. Ding, Y. G. Xu, J. Fu, and M. Q. Kuang, Theoretical studies of the EPR spectra and local structures for Cu²⁺ and V⁴⁺ doped halo-borate glasses, *Mater. Sci. Eng. B-Adv*, **271** (2021) 115245, <https://doi.org/10.1016/j.mseb.2021.115245>
34. W. L. Feng, and J. Y. Xue, Calculations with the two-mechanism model for the spin-Hamiltonian parameters of Mo⁵⁺ ions in phosphate glasses, *J. Non-Cryst. Solids.*, **385** (2014) 160, <https://doi.org/10.1016/j.jnoncrysol.2013.11.028>
35. J. Z. Lin, and B. F. Zhang, Theoretical studies on the electron paramagnetic resonance parameters for the tetragonal VO₂⁺ center in CaO-Al₂O₃-SiO₂ system, *Rev. Mex. Fis.* **64** (2018) 13, <https://doi.org/10.31349/RevMexFis.64.13>
36. R. M. Peng, Y. Mei, W. C. Zheng, and C. F. Wei, Assignments of the optical band positions and theoretical calculations of the spin-Hamiltonian parameters for the tetragonal W⁵⁺ octahedral clusters in tungsten oxide (WO₃)-based glasses, *Physica B*, **461** (2015) 106, <https://doi.org/10.1016/j.physb.2014.12.018>
37. S. Y. Wu, H. M. Zhang, P. Xu, and S. X. Zhang, Studies on the defect structure for Cu₂⁺ in CdSe nanocrystals, *Spectrochim. Acta. A.*, **75** (2010) 230, <https://doi.org/10.1016/j.saa.2009.10.016>
38. C. C. Ding, S. Y. Wu, L. J. Zhang, X. F. Hu, and G. L. Li, Studies of the local distortions for nd¹ (Mo⁵⁺ and W⁵⁺) impurities in KTP crystals, *Polyhedron.*, **105** (2016), 71, <https://doi.org/10.1016/j.poly.2015.12.013>
39. Y. Mei and W. C. Zheng, Spin-Hamiltonian parameters and tetragonal distortion for the (WO₆)⁷⁻ octahedral centers in the WO₃-doped Zn(PO₄)₂ ZnO nanopowders, *Optik*, **179** (2019) 965, <https://doi.org/10.1016/j.ijleo.2018.11.055>
40. Y. D. Li, B. J. Chen, H. M. Zhang and C. D. Feng, Theoretical study of the local distortions and the EPR parameters for VO₂⁺ and Cu₂⁺ centers in oxyborate glasses, *J. Lumin.* **234** (2021) 117973, <https://doi.org/10.1016/j.jlumin.2021.117973>
41. A. Bals and J. Kliava, Simulations of EPR spectra for d¹ ions with distributed spin hamiltonian parameters, *J. Magn. Reson.*, **53** (1983) 243, [https://doi.org/10.1016/0022-2364\(83\)90030-6](https://doi.org/10.1016/0022-2364(83)90030-6)
42. X. M. Zhang, L. X. Yang, B. Q. Zen, and W. L. Yu, Spectroscopic properties of Cr³⁺ ions at the defect sites in cubic uoroperovskite crystals, *Phys. Rev. B.*, **50** (1994) 6756, <https://doi.org/10.1103/PhysRevB.50.6756>

43. J. R. Lyu, S. Y. Wu, Y. J. Hong, H. Wu, and H. N. Dong, Theoretical studies of the concentration dependences of d-d transition band, g factors, and local distortion for V⁴⁺ in Li₂O-PbO-B₂O₃-P₂O₅ glasses, *Magn. Reson. Chem.*, **60** (2022) 398, <https://doi.org/10.1002/mrc.5232>
44. W. L. Feng, X. M. Li, W. C. Zheng, Y. G. Yang, and W. Q. Yang, EPR g factors and tetragonal distortion for the isoelectronic Ni⁺ and Cu²⁺ centers in the CuGaSe₂ crystal, *J. Magn. Mater.*, **323** (2011) 528, <https://doi.org/10.1016/j.jmmm.2010.10.003>
45. X. S. Liu, S. Y. Wu, G. Q. Shen, X. Y. Yu, Q. S. Zhu, and X. Y. Li, Theoretical analysis on lead halides content dependences of defect structures and spin Hamiltonian parameters in lead halo borate glass composites with V₂O₅ dopants, *Compos. Part. A-Appl. S.*, **146** (2021) 106414, <https://doi.org/10.1016/j.compositesa.2021.106414>
46. Y. Mei, X. X. Wu, W. L. F., and W. C. Zheng, Investigation of the EPR Parameters and Local Structures for the Tetragonal Ti₃⁺-O₂²⁻-VM Centers in MgO and CaO Crystals, *Acta. Phys. Pol. A.*, **134** (2018) 481, <https://doi.org/10.12693/APhysPolA.134.481>
47. L. X. Zhu and M. J. Wang, Investigations of the Defect Structures and g Factors for SnO₂:Cr⁵⁺, SnO₂:V⁴⁺, and TiO₂:V⁴⁺ Crystals, *Appl. Magn. Reson.*, **47** (2016) 247, <https://doi.org/10.1007/s00723-015-0754-4>
48. L. X. Zhu and M. J. Wang, Investigations of g-factor for Ti³⁺, V⁴⁺ and Cr⁵⁺ in anatase, *Dalton. T.*, **42** (2013) 16289, <https://doi.org/10.1039/C3DT51672E>
49. W. L. Feng and W. C. Zheng, Theoretical explanation of spin-Hamiltonian parameters for the rhombic Mo⁵⁺ octahedral clusters in molybdenum phosphate glasses, *Mol. Phys.*, **113** (2015) 3228, <https://doi.org/10.1080/00268976.2015.1014005>