

# Raman spectroscopy and electrical properties of polypyrrole doped dodecylbenzene sulfonic acid/ $Y_2O_3$ composites

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The doped dodecylbenzene sulfonic acid (DBSA) with polypyrrole (PPy) and also incorporated an increasing concentration of  $Y_2O_3$  to obtain the composites of PPy-DBSA- $Y_2O_3$  via chemical polymerization technique. The PPy-DBSA- $Y_2O_3$  composites formation were confirmed by interaction between PPy-DBSA and  $Y_2O_3$ -particles utilizing Raman spectroscopy. The SEM micrographs show that the composites are in the form of lengthened chains; increase in the particles size as equated with pristine PPy and  $Y_2O_3$  was also studied. Thermal stability of PPy-DBSA-  $Y_2O_3$  composites was improved as enhanced the load of  $Y_2O_3$ -particles. The increase in DC conductivity by mixing  $Y_2O_3$  into PPy-DBSA at all temperatures showed the three-dimensional Mott's variable range hopping model. Density of localized states, hopping dimension as well as activation energy are computed and found to be affected due to the presence of  $Y_2O_3$  in DBSA-PPy. The ESR of  $Y_2O_3$  ( $\sim 12 \Omega$ ), PPy ( $\sim 11.80 \Omega$ ), PPy-DBSA ( $\sim 11.30 \Omega$ ) and PPy-DBSA-8% $Y_2O_3$  composite ( $\sim 9.50 \Omega$ ). EIS results confirm that the PPy-DBSA-8%  $Y_2O_3$  composite with a low value of impedance gives a maximum value of electrical conductivity.

**Keywords:** Raman analysis; EIS; PPy-DBSA;  $Y_2O_3$ ; DC Conductivity.

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## 1. Introduction

Conjugated polymers have been a significant research area in pure and applied fields since the last two years [1-4]. The conjugated polymer, for example polypyrrole, polyaniline, etc., are polymers with electronically conjugated backbones that, when doped, conduct electricity.

These materials have recently fascinated attention due to their possible use in numerous technological applications that could benefit from their exceptional polymeric and electronic properties [5]. The possible applications of conjugated polymers are the fabrication of solid-state devices, for instance, schottky diodes, microelectronics as well as solar cells. Numerous conjugated polymers possess poor environmental stability and low mechanical properties, owing to which they were not valuable for the fabrication of devices. To minimize these difficulties, many efforts were made to synthesize the conjugated polymer composites to achieve the mechanical properties [6-8]. Some of the variations required to synthesize the hybrid materials provide an organic material through inorganic oxides of various metals, for example,  $SnO_2$  [9],  $TiO_2$  [10], fly ash [11],  $Fe_3O_4$  [12] and  $ZrO_2$  [13] are also mixed conjugated polymers to provides the composites. The study of charge transport mechanism in these composites was a significant part of polymer investigation. Even though a combination of functional dopant like dodecylbenzene sulfonic acid (DBSA) is added into the conjugated polymer, which may distort the polymer chain and produces polarons and bipolarons [14]. Conjugated polymers are very cheap organic semiconductor material having polarons and

bipolarons moving charge carriers that are accountable for their electrical conduction and also suitable for the fabrication of electronic devices [15]. The charge transport mechanism taking place in polyacetylene was observed due to the moving solitons [16]. Among various conjugated polymers, the polypyrrole (PPy) is frequently used owing to elevated electrical conductivity and good environmental stability [17]. Among lots of inorganic material,  $Y_2O_3$  has been broadly studied due to its good thermal stability and is mainly used in phosphors host matrices [18] and dielectric insulator of electro-luminescent devices [19-20]. Moreover,  $Y_2O_3$  is a very essential ceramic compound that can be used as a corrosion resistance material, temperature dependent electronic devices fabrication particularly due to high thermal stability. The  $Y_2O_3$  plays very important role in laser host materials to enhance the life span of luminous lamps and their thermal stability [21-22]. Finally, the  $Y_2O_3$  owing to high surface area was utilized as a catalyst carrier [23] in order to increase the conductivity of temperature dependent electronic devices. Until now, a literature review refers to temperature dependent DC conductivity, thermal stability and impedance investigations on PPy-DBSA- $Y_2O_3$  are unusual. In present work, we report chemical preparation of PPy-DBSA- $Y_2O_3$  composites, and the prepared material is characterized using Raman spectroscopy, TGA and EIS analysis.

The results cover thermal stability, impedance and temperature dependent DC conductivity; hopping length in addition to activation energy were calculated and discussed the obtained results were valuable for electronic devices fabrication.

## 2. Experimental details

### 2.1. Chemicals and material

Pyrrole (Fluka) was refined under reduced pressure and temperature preceding to its usage. Ammonium persulphate, Dodecylbenzene sulfonic acid in addition to Yttrium oxide were purchased (Sigma-Aldrich) utilized as achieved. The whole material was employed as presented without slightly further purification.

### 2.2. Preparation of PPy and PPy-DBSA- $Y_2O_3$

The 0.3 mol of Pyrrole was incorporated gradually into the solution which was placed on a magnetic stirrer. While strong stirrers retain the pH value between 0 and 1, 30% HCl was also added into the reaction solution. In this solution, the 0.15 mol of DBSA and  $Y_2O_3$  powder were both dispersed into a reaction mixture under a strong stirrer. In the next hour, the required quantity of ammonium persulphate was dispersed in 100-mL distilled water dropwise under a magnetic stirrer 1/2-hour period. The molar ratio of dopant/ monomer/oxidant was kept 1/4:1:1. At that time, one-liter methanol was mixed into the solution mixture which was kept at 30°C for 48 hours to obtain the entire polymerization. Finally, the solution is cleaned and purified using de-ionized water until the filtrate solution becomes colorless. The greenish-black paste of polypyrrole and PPy-DBSA- $Y_2O_3$  was achieved by being dehydrated in a vacuum oven at 70°C for 24 hours.

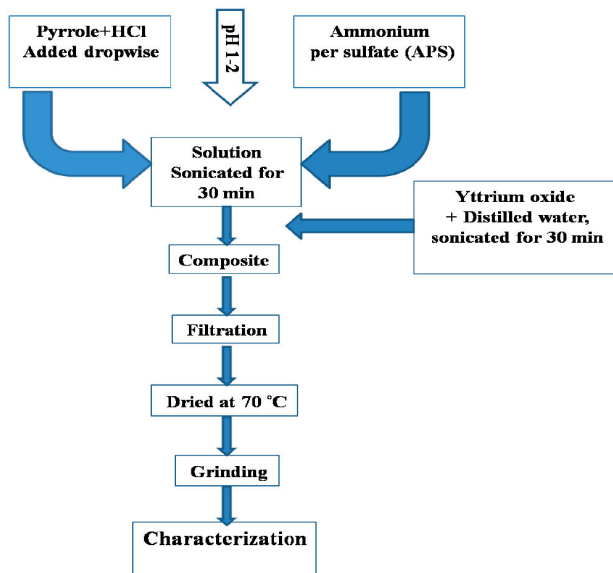


FIGURE 1. Flow chart for the preparation of PPy-DBSA- $Y_2O_3$  composites by in-situ polymerization process.

## 3. Measurements

A Renishaw RM 1000 (He-Ne laser) Olympus metallurgical microscope and a CCD detector were utilized for all samples.

Scanning electron microscope was carried out on EVO50 ZEISS tool. The thermogravimetric analysis (TGA) was carried out on Mettler thermo balance STAR S.W. 8.10 for all samples. Temperature-dependent DC conductivity for entire samples was executed by Keithley, 2400 electrometers along with the current source electrometer, employing the two-probe method. Electrochemical impedance spectroscopy (EIS) measurements are taken at 0.7 V potential with 0.005 Vrms amplitude in the ( $10^5$  to 1) Hz frequency range.

## 4. Results and discussion

### 4.1. Raman spectroscopy analysis

Raman spectroscopy was employed to observe structural variations by adding  $Y_2O_3$ -particles into DBA-PPy. Figures 2 (a-b) shows the Raman spectra of polypyrrole and DBSA-PPy. The peaks were observed at  $1055\text{ cm}^{-1}$  ( $1060\text{ cm}^{-1}$  for polypyrrole) associated with the polaron quinonoid structure. Moreover, two peaks were observed at  $934\text{ cm}^{-1}$  and  $1255\text{ cm}^{-1}$  also correlated to bipolaron quinonoid structure that confirmed the doped PPy structure [24]. The peak observed at  $1578\text{ cm}^{-1}$  is a typical quinoid C = C stretched mode of polymer chain. The DBSA-PPy composite with improved bipolarons ( $1255\text{ cm}^{-1}$ ) as equated to polarons in PPy spectra also increased the electrical conductivity [25].

Figures 2 (c-e) shows the Raman spectra of PPy-DBSA- $Y_2O_3$  by incorporating the various load ratio of  $Y_2O_3$  from 2%, to 8% into PPy-DBSA. The major peaks were observed

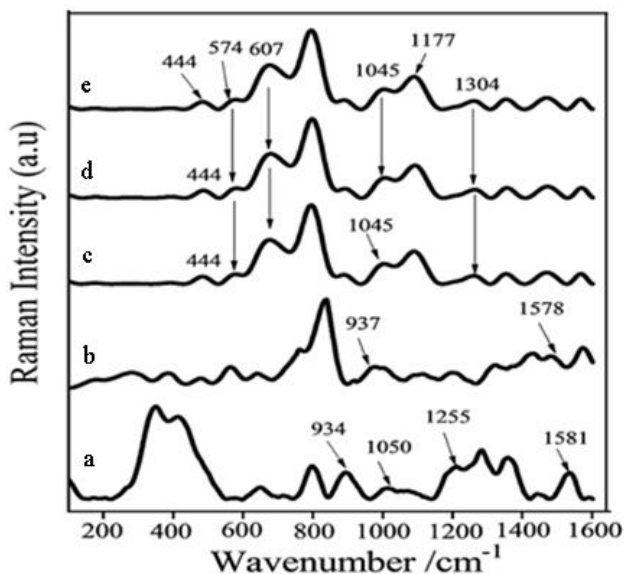


FIGURE 2. Raman spectra of a) pure PPy b) PPy-DBSA c) 2% d) 4% e) 8% load of  $Y_2O_3$  in PPy-DBSA.

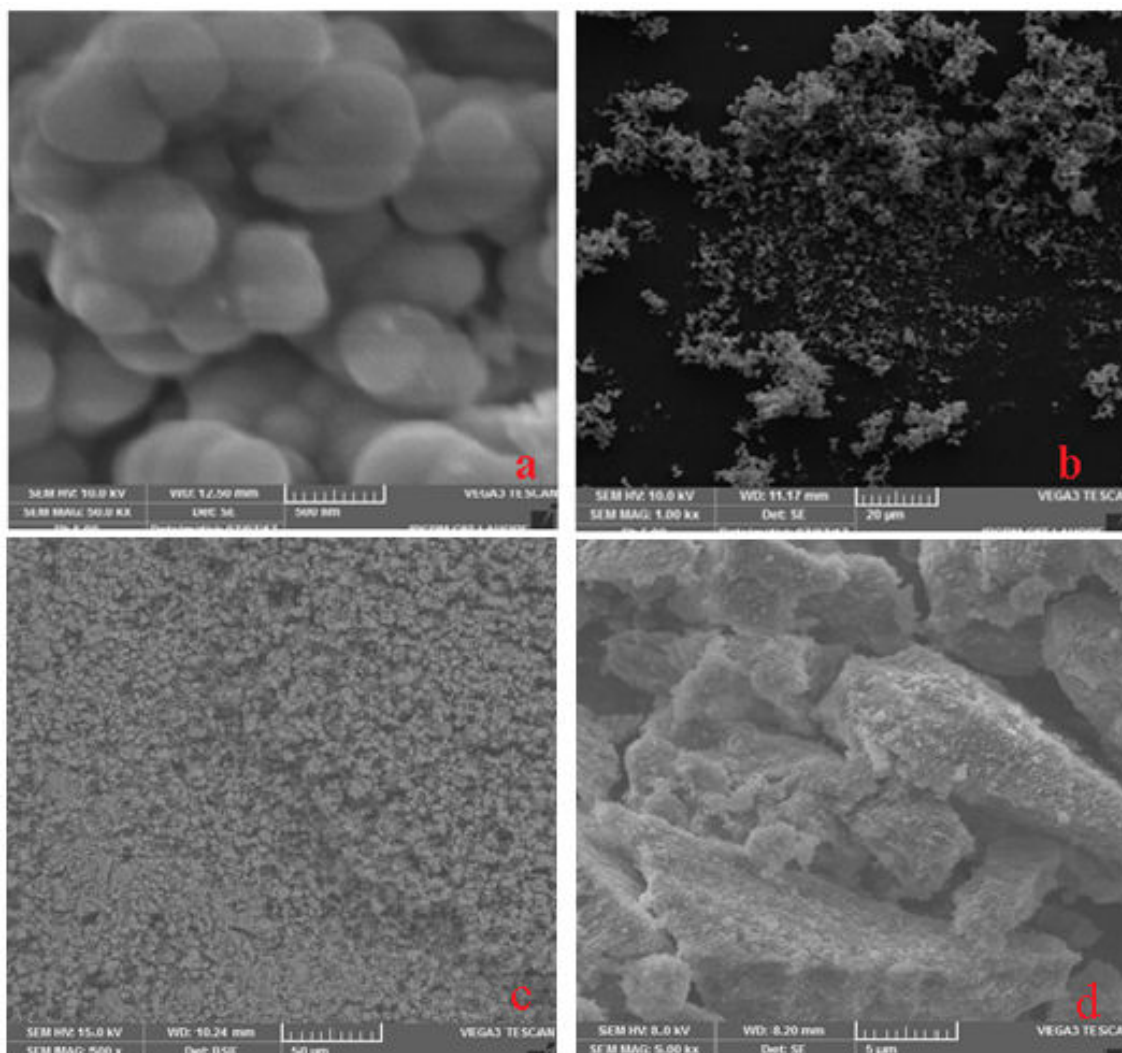


FIGURE 3. Scanning electron micrographs of a) pure PPy b) PPy-DBSA c) PPy-DBSA-8%  $Y_2O_3$  composite d)  $Y_2O_3$ .

in PPy-DBSA- $Y_2O_3$  composites at  $1177\text{ cm}^{-1}$  (N-C extending band),  $1304\text{ cm}^{-1}$  (C-N extending band),  $1045\text{ cm}^{-1}$  (C-H in-plane distortion) as well as pristine  $Y_2O_3$  peak was seen at  $444\text{ cm}^{-1}$ . Furthermore, two peaks were also examined at  $607\text{ cm}^{-1}$  correlated with distortion in the benzene ring [26] and at  $574\text{ cm}^{-1}$  attributed to a cross-link between PPy chains [27]. The peak observed at  $607\text{ cm}^{-1}$  was identified as distortion of a benzene ring in PPy backbone. Figure 1(e) shows that the intensity of the peak at  $607\text{ cm}^{-1}$  increases as compared to the peak observed at  $574\text{ cm}^{-1}$  which suggests the confirmation regarding the interactions between the polymer elements. It has been revealed that after mixing the  $Y_2O_3$  presenting the strong interaction between PPy chains was also observed [28]. The comparative intensity of a peak investigated at  $607\text{ cm}^{-1}$  shows that PPy-DBSA composite including  $Y_2O_3$ -nanoparticles provides a strong interchain interaction. This effect was also observed in the composite that contained 8% load of  $Y_2O_3$ . From this investigation, it can be concluded that the induced variations in the observed spectrum of new peaks in all composites confirm the forma-

tion of strong interaction between the polymeric chain and  $Y_2O_3$ -nanoparticle.

#### 4.2. Scanning electron microscopy (SEM) analysis

Figures 3(a-d) exhibits SEM micrographs of PPy, PPy-DBSA and its PPy-DBSA- $Y_2O_3$  composite as well as pure  $Y_2O_3$ -particles. It was clearly observed from the SEM image of polypyrrole (PPy) that it has bunches of globular shaped particles. The SEM morphology of PPy doped DBSA also shows the cluster of particles which might be due to better inter-chain interaction and consequences to enhance the conductivity [29]. The SEM image is suggestive of the hemispherical nature of polymer as clusters in PPy-DBSA- $Y_2O_3$  composite and flakier structure of pristine  $Y_2O_3$ . The  $Y_2O_3$ -particles are encircled into PPy-DBSA chain owing to robust particle-interaction. From this, it may be concluded that PPy-DBSA- $Y_2O_3$  composite is seeing more progress in particle size and a platelet structure.

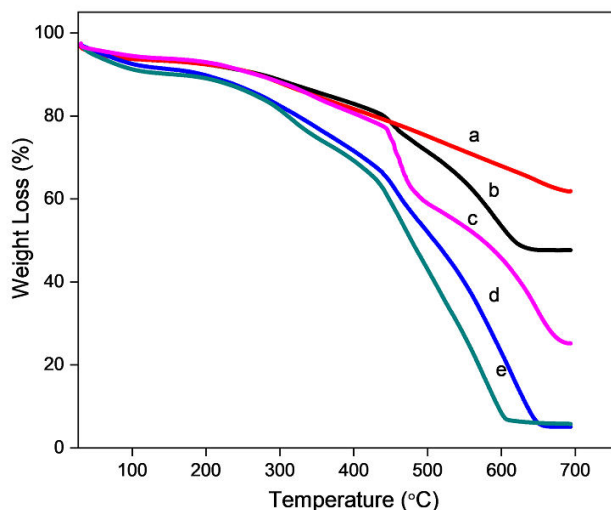


FIGURE 4. TGA of a) PPy b) DBSA-PPy c) 2% d) 4% e) 8% load of  $Y_2O_3$  in PPy-DBSA.

#### 4.3. Thermogravimetric analysis (TGA)

Figures 4(a-e) shows the typical TGA curves for PPy, DBSA-PPy and their composites with 2% to 8% weight ratio of  $Y_2O_3$  nanoparticles doped in PPy-DBSA. The TGA study shows the two-phase of mass loss: the initial phase ( $\sim 100^\circ C$ ) can be permitted humidity desorption from polymer; next phase starts from  $230^\circ C$  to  $550^\circ C$  and it is owing to thermal degradation of the polymer [30]. In comparison to DBSA-PPy, it might be seen that there is no sharp inception of thermal breakdown that occurs at  $300^\circ C$  plotted in the Fig. 4 curve (b). From Fig. 4 curve (e), it can be observed that the mass loss arises in DBSA-PPy-8% $Y_2O_3$  slowly as compared to pure PPy owing to volatilization of small molecules. Hence, the  $Y_2O_3$  mixed into DBSA-PPy improves thermal stability of all the composites [31].

#### 4.4. Electrochemical impedance analysis

At room temperature, the electrochemical impedance was measured using a 1M-KOH electrolyte and the mass of the sample was chosen (0.01 g). The Nyquist plot of all samples has the vertical line in the high-frequency region, although a semicircle executed in the low-frequency region with a low value of impedance indicates the capacitive nature of all samples [32]. The impedance was measured for all samples with equivalent series resistance (ESR). The ESR of PPy-DBSA-8%  $Y_2O_3$  composite is lower as compared to pure  $Y_2O_3$ , PPy, and PPy-DBSA as executed in Figs. 5(a-d). ESR of  $Y_2O_3$  ( $\sim 12 \Omega$ ), PPy ( $\sim 11.80 \Omega$ ), PPy-DBSA ( $\sim 11.30 \Omega$ ) and PPy-DBSA-8%  $Y_2O_3$  composite ( $\sim 9.50 \Omega$ ).

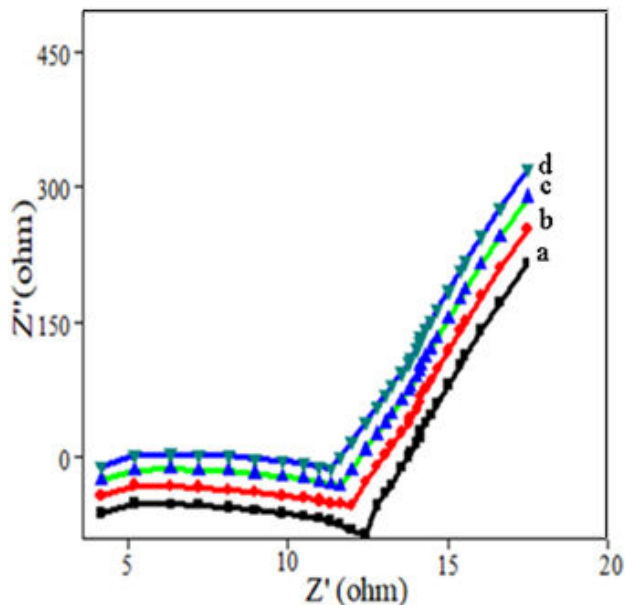


FIGURE 5. Nyquist graphs of a)  $Y_2O_3$  b) PPy c) DBSA-PPy and d) PPy-DBSA-8%  $Y_2O_3$ , in 5 mV AC

#### 4.5. Electrical properties

##### 4.5.1. DC conductivity

A series of composites was synthesized to maintain the concentrations of pyrrole-constant but adding an increasing extent of  $Y_2O_3$ -nanoparticles into DBSA-PPy. Composites with an increasing quantity of  $Y_2O_3$  produce different conductivity as shown in Fig. 6. The increase in conductivity via the enhancing load of  $Y_2O_3$  can be attributed to decreasing the conductive path because of  $Y_2O_3$  as an insulator nature. The increasing quantity of  $Y_2O_3$  reduces the level of conjugated  $\pi$ -bonds in PPy-DBSA and disorders the polymer chains, which increases the conductivity of composites.

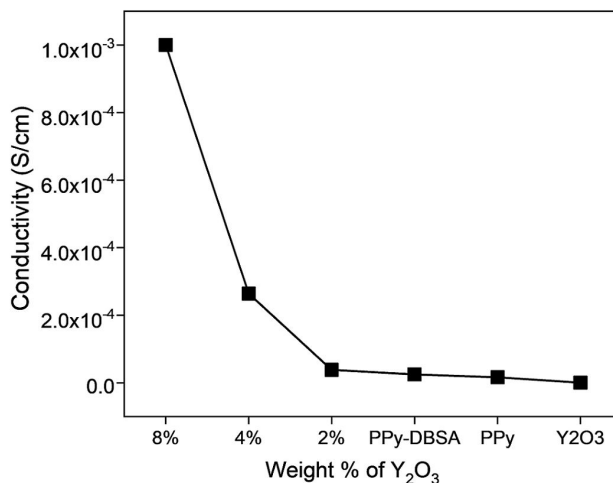


FIGURE 6. Conductivities of PPy-DBSA $Y_2O_3$  composites with different weight % of  $Y_2O_3$ .

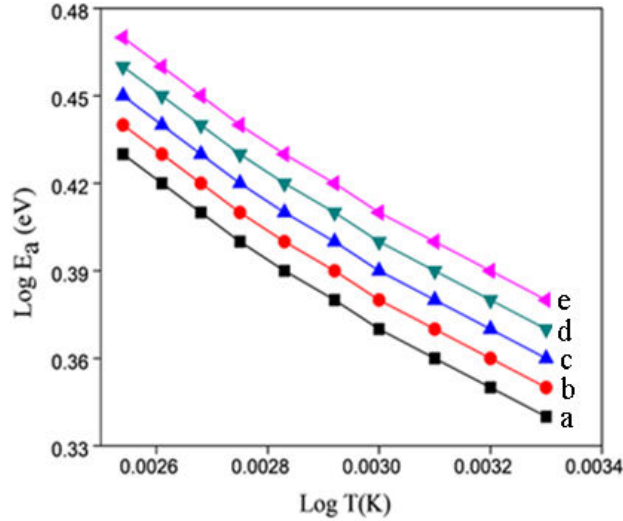


FIGURE 7. Graph of  $\text{Log}E_a$  against  $\text{Log}T$  for a) PPy b) PPy-DBSA c) 2% d) 4% e) 8% load of  $\text{Y}_2\text{O}_3$  in PPy-DBSA composites.

Consequently,  $\text{Y}_2\text{O}_3$ -particles perform an important role in the conductivity of composites.

#### 4.5.2. Temperature dependant DC conductivity

The correlation between DC conductivity and temperature for polymer samples may offer significant information about the charge transport mechanism through the polymer scheme. Mott's variable range hopping model adopts the following equations [33-35].

$$\sigma = \sigma_0 \exp(T_o/T)^{\frac{1}{1+n}} (\text{Scm}^{-1}), \quad (1)$$

where  $n$  the above expression,  $\sigma_0 (\text{Scm}^{-1})$  pre-exponential factor is the conductivity of material at the specified temperature  $T(\text{K})$ , despite  $T_o(\text{K})$  Mott's characteristic temperature. The value of  $\sigma$  could be attained from intercepts in addition to slope of  $\text{Log}(\sigma)$  plot against  $T^{-1/4}$ . The following equations can be adopted to calculate the average hopping length  $[R, \text{cm}]$ , density of states  $[N(E_F), \text{cm}^{-3}\text{eV}^{-1}]$  as well as hopping activation energy  $[W, \text{eV}]$  as:

$$\sigma_0 = e^2 R^2 \nu_{ph} N (E_F), \quad (2)$$

$$T_o = \frac{\lambda \alpha^3}{kN(E_F)} \quad (\text{k}), \quad (3)$$

$$R = \left[ \frac{9}{8\pi \alpha k T N(E_F)} \right]^{\frac{1}{4}} \quad (\text{cm}), \quad (4)$$

$$W = \frac{3}{4\pi R^3 N(E_F)} \quad (\text{eV}). \quad (5)$$

The charge transport mechanism is determined by employing the data in terms of Mott's variable range hopping model. From Eq. (1) exponent  $n$  is a dimensionality system, its value  $n = 1, 2$  or  $3$  showing 1, 2 or 3-dimensional variable range hopping charge transport mechanism.

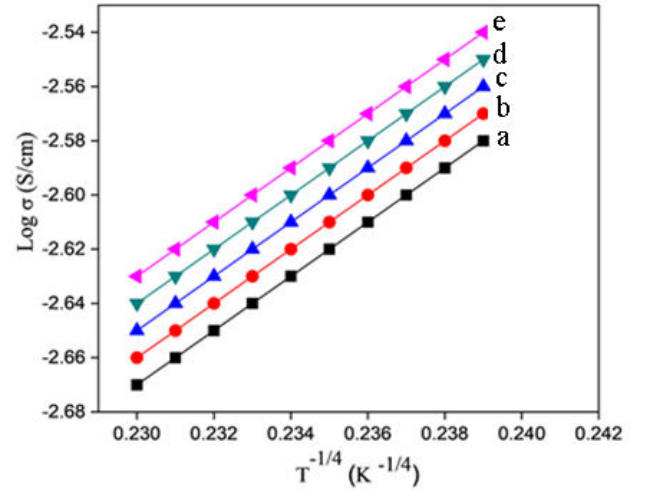


FIGURE 8. Variation in DC conductivity with Temperature for a) PPy b) PPy-DBSA c) 2% d) 4% e) 8% load of  $\text{Y}_2\text{O}_3$  in PPy-DBSA.

The activation energy  $E_a$  is given by Eq. (6):

$$E_a = d(\text{Log}\sigma)/d(1/KT). \quad (6)$$

Therefore, activation energy is calculated from the slope of  $\text{Log}(\sigma)$  vs.  $1/T$  and distorted in finding the hopping charge transport mechanism. By using Eqs. (1) and (6), we get the following Eq. (7):

$$E_a = \gamma K T_o \left( \frac{T_o}{T} \right)^{\gamma-1} \quad (7)$$

where  $\gamma = 1/(1+n)$ .

By plotting  $\text{Log}E_a$  vs  $\log T$  (Fig. 7) slope of a straight line  $-(\gamma-1)$  equal to  $3/4$  is attained as the hopping exponent may be  $\gamma \sim 1/4$  in addition to  $n \sim 3$ . This confirmed that the 3-dimensional variable range hopping charge transport mechanism dominates in PPy-DBSA- $\text{Y}_2\text{O}_3$  composites, similar to the 3-dimensional variable range hopping charge transport model to pretend in the case of pure PPy. Hence,  $\text{Log}(\sigma)$  has a direct relationship with  $T^{-1/4}$  as shown in Fig. 8. The density of localized states decreases whereas hopping length, activation energy as well as  $\sigma_{dc}$  were observed to be increased by the raised load ratio of  $\text{Y}_2\text{O}_3$ . The effective dimensionality  $n$  depends upon the interchain coupling of the charge transport mechanism. It can be observed that the charge transport mechanism in conjugated polymers rises via a 3-dimensional variable range hopping model network. The conjugated polymers with various morphology at different temperatures show conductivity data with exponent  $n$  either equal to  $1/4$  [36-38] or  $1/2$  [39-41] strongly depending on the temperature. The value of the exponent  $n$  for temperature dependent DC conductivity is  $1/4$  and effective dimensionality  $n = 3$  [ $\gamma = 1/(1+n)$ ]. Consequently, conductivity data leads to a 3-dimensional variable range hopping model for strong interchain coupling. The value of the exponent  $n$  is  $1/2$  ( $n = 1$ ) and was described in terms of the

TABLE I. Mott's parameters of PPy, PPy-DBSA and PPy-DBSA-Y<sub>2</sub>O<sub>3</sub> composites at 303 K.

Samples	T <sub>0</sub> (K)	N(E <sub>F</sub> ) (cm <sup>-3</sup> eV <sup>-1</sup> )	R (cm)	W <sub>hop</sub> (eV)	σ <sub>dc</sub> (S/cm)
PPy	2.43 × 10 <sup>11</sup>	3.20 × 10 <sup>25</sup>	5.90 × 10 <sup>-9</sup>	0.458	2.30 × 10 <sup>-4</sup>
PPy-DBSA	3.26 × 10 <sup>11</sup>	2.40 × 10 <sup>25</sup>	6.40 × 10 <sup>-9</sup>	0.450	2.64 × 10 <sup>-4</sup>
PPy-DBSA-2% Y <sub>2</sub> O <sub>3</sub>	5.40 × 10 <sup>11</sup>	1.50 × 10 <sup>25</sup>	7.30 × 10 <sup>-9</sup>	0.416	2.76 × 10 <sup>-3</sup>
PPy-DBSA-4% Y <sub>2</sub> O <sub>3</sub>	4.30 × 10 <sup>11</sup>	1.80 × 10 <sup>25</sup>	6.90 × 10 <sup>-9</sup>	0.427	2.83 × 10 <sup>-3</sup>
PPy-DBSA-8% Y <sub>2</sub> O <sub>3</sub>	3.60 × 10 <sup>11</sup>	2.20 × 10 <sup>25</sup>	6.60 × 10 <sup>-9</sup>	0.436	3.22 × 10 <sup>-3</sup>

Efros-Shklovskii [41] model or granular metal model [42] variable range hopping in the case of weak interchain coupling. The best fit conductivity data to our synthesized PPy-DBSA-8% Y<sub>2</sub>O<sub>3</sub> sample follows the 3-dimensional variable range hopping model. From these outcomes, we found that robust interchain interaction exists in PPy-DBSA-Y<sub>2</sub>O<sub>3</sub> composites. However, the reduction in DC conductivity was investigated owing to the insulator nature of Y<sub>2</sub>O<sub>3</sub> into DBSA-PPy.

## 5. Conclusions

PPy-DBSA and also mixed Y<sub>2</sub>O<sub>3</sub> nanoparticles have been successfully prepared PPy-DBSA-Y<sub>2</sub>O<sub>3</sub> composites utilizing the chemical polymerization route. The structure of prepared samples was explored with Raman spectroscopy. The

SEM also confirms the platelet structure in PPy-DBSA-Y<sub>2</sub>O<sub>3</sub> composites. The TGA curves confirmed that the thermal stability is increased in all composites by incorporation of the Y<sub>2</sub>O<sub>3</sub>-particles rather than the pure polymer. The insulator behaviour of Y<sub>2</sub>O<sub>3</sub> disrupts delocalization of charge carriers and shrinks the inter-chain coupling, due to which σ<sub>dc</sub> = 3.20 × 10<sup>-3</sup> S/cm at 30°C for PPy-DBSA-8% Y<sub>2</sub>O<sub>3</sub> is greater as compared to σ<sub>dc</sub> = 2.70 × 10<sup>-3</sup> S/cm at 3030°C for PPy-DBSA. The temperature dependent DC conductivity was investigated and the conduction mechanism in these samples follows 3-dimensional variable range hopping model. The ESR of Y<sub>2</sub>O<sub>3</sub> (~12 Ω), PPy (~11.80 Ω), PPy-DBSA (~11.30 Ω) and PPy-DBSA-8% Y<sub>2</sub>O<sub>3</sub> composite (~9.50 Ω). Hence, the calculated values of density of localized states, hopping length as well as activation energy are observed to be varied by mixing Y<sub>2</sub>O<sub>3</sub>-particles into DBSA-PPy.

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